



THE UNIVERSITY *of* EDINBURGH

Edinburgh Research Explorer

Dimensional crossover of correlated anion disorder in oxynitride perovskites

Citation for published version:

Johnston, H, Black, AP, Kayser, P, Oró-solé, J, Keen, DA, Fuertes, A & Attfield, P 2018, 'Dimensional crossover of correlated anion disorder in oxynitride perovskites', *Chemical Communications*.
<https://doi.org/10.1039/C8CC03462A>

Digital Object Identifier (DOI):

[10.1039/C8CC03462A](https://doi.org/10.1039/C8CC03462A)

Link:

[Link to publication record in Edinburgh Research Explorer](#)

Document Version:

Peer reviewed version

Published In:

Chemical Communications

General rights

Copyright for the publications made accessible via the Edinburgh Research Explorer is retained by the author(s) and / or other copyright owners and it is a condition of accessing these publications that users recognise and abide by the legal requirements associated with these rights.

Take down policy

The University of Edinburgh has made every reasonable effort to ensure that Edinburgh Research Explorer content complies with UK legislation. If you believe that the public display of this file breaches copyright please contact openaccess@ed.ac.uk providing details, and we will remove access to the work immediately and investigate your claim.



Dimensional crossover of correlated anion disorder in oxynitride perovskites

Hannah Johnston,^a Ashley P. Black,^b Paula Kayser Gonzalez,^a Judith Oró-Solé,^b David A. Keen,^c Amparo Fuertes,^{*b} and J. Paul Attfield^{*a}

Received 00th January 20xx,
Accepted 00th January 20xx

DOI: 10.1039/x0xx00000x

www.rsc.org/

A simple crossover from two-dimensional to three-dimensional correlated disorder of O and N atoms on a cubic lattice has been discovered within the $\text{Ba}_{1-x}\text{Sr}_x\text{TaO}_2\text{N}$ series of perovskite oxynitrides. The crossover is driven by lattice expansion as x decreases, and provides a rapid increase in entropy due to a change from subextensive to extensive configurational entropy regimes.

Correlated disorders of atoms or magnetic moments, where local structure rules do not give rise to long range (crystallographic) order, have been identified in many crystalline materials such as water and spin ices, metal cyanides and molecular systems.¹ Both two-dimensional (2D) and three-dimensional (3D) examples are known, but dimensional crossovers which are associated with changes of chemical or physical properties as new degrees of freedom become available are rare for correlated disorder. Dipolar spin-ice correlations in the pyrochlore $\text{Dy}_2\text{Ti}_2\text{O}_7$ were switched from 3D to 2D behaviour through application of a magnetic field,² but analogous atomic dimensional crossovers are not reported. Here we describe a simple strain-driven 2D to 3D crossover on a cubic lattice of correlated disorder of O and N atoms within the $\text{Ba}_{1-x}\text{Sr}_x\text{TaO}_2\text{N}$ perovskites series.

AMO_2N or AMON_2 oxynitride perovskites have useful properties such as water-splitting photocatalysis and high dielectric constants.^{3,4} 2D correlations of O/N atoms have been evidenced in many of these materials;⁵ from neutron and electron diffraction studies of SrMO_2N ($\text{M} = \text{Nb}, \text{Ta}$),⁶ LaTaON_2 ,⁷ and RVO_2N ($\text{R} = \text{Pr}, \text{Nd}$)^{8,9} perovskites, and from electron diffraction studies of EuMO_2N ($\text{M} = \text{Nb}, \text{Ta}$)¹⁰ and EuWON_2 .¹¹ Anions within the AMO_2N perovskites are locally ordered as MO_4N_2 octahedra

where the two nitrides adopt a *cis* (90°) configuration which is favoured by the higher covalency of the M-N bond (Fig. 1a).¹² Combining the *cis* configuration with the linear coordination of each nitride by two M cations results in the formation of zig zag –M-N– chains and rings that spontaneously segregate into 2D planes within the perovskite structure, as shown in Fig. 1d. This is evidenced by a partial segregation of anions within the average unit cell (Fig. 1b) that is observable by neutron diffraction due to high O/N scattering contrast. A chemical symmetry between the anion orders of AMO_2N and AMON_2 perovskites obtains from interchanging O and N positions so that 2D layers of *cis*-TaO chains are found in LaTaON_2 .

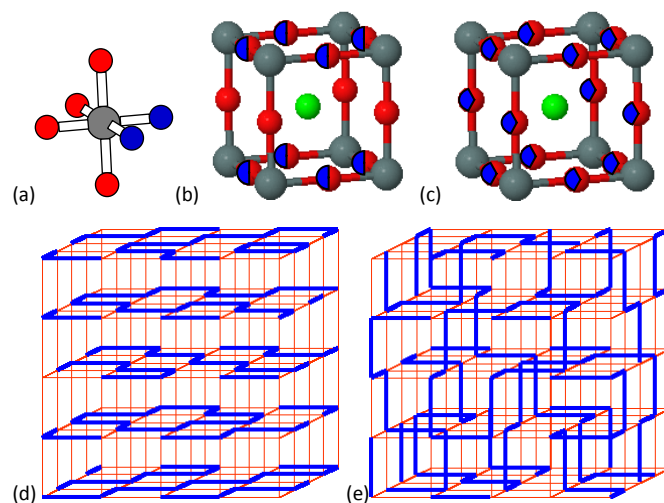


Figure 1 Anion orders in AMO_2N perovskites, with N/O atoms shown as blue/red throughout. (a) Local *cis* (90°) coordination of N atoms in all MO_4N_2 octahedra. (b) The average anion distribution in 2D materials which have a tetragonal $P4/mmm$ cell with two anion sites occupied 50% by O and 50% by N, and one site of 100% O. (c) The average anion distribution in 3D materials which have the cubic $Pm\bar{3}m$ perovskite structure with 67% occupancy of O and 33% N at all anion sites. (d) Local picture of *cis*-MN chains confined to 2D, where thick blue/thin red lines correspond to M-N-M/M-O-M connections, and the average site occupancies correspond to those in (b). (e) Local picture of *cis*-MN chains propagating in 3D, giving the average site occupancies shown in (c). Correlated anion disorder in

^a CSEC and School of Chemistry, University of Edinburgh, King's Buildings, Mayfield Road, Edinburgh, EH9 3JZ, U.K.

^b Institut de Ciència de Materials de Barcelona (ICMAB-CSIC), Campus UAB, 08193 Bellaterra, Spain.

^c ISIS Facility, Rutherford Appleton Laboratory, Didcot, OX11 0QX, U.K.

Electronic Supplementary Information (ESI) available: N analysis, electron microscopy and further refinement results. See DOI: 10.1039/x0xx00000x

Open data for this paper are at <https://dashshare.is.ed.ac.uk/handle/10283/838>

AMON₂ perovskites is chemically symmetric through reversal of O and N in all figures.

An alternative 3D distribution of the *cis*-chains in oxynitride perovskites is also possible (Fig. 1e), but no evidence for a 2D to 3D crossover was found in high temperature experiments. Chains were found to be confined to 2D up to the highest measured temperature of 1100 °C in neutron diffraction studies of SrTaO₂N and LaTaO₂N.⁷ However, while a neutron diffraction and PDF (pair distribution function) study of BaTaO₂N reported that *cis*-TaO₄N₂ octahedra were present, it was not clear whether N atoms are distributed in 2D or 3D.¹³ We have therefore explored the full series of solid solutions Ba_{1-x}Sr_xTaO₂N (0 ≤ x ≤ 1) using high resolution neutron and x-ray diffraction to establish the 2D/3D nature and evolution of the correlated anion chain disorder.

Highly crystalline powder samples of the solid solutions Ba_{1-x}Sr_xTaO₂N (x = 0, 0.2, 0.4, 0.6, 0.8 and 1) were prepared using stoichiometric amounts of BaCO₃ (Alfa Aesar 99.997 %), SrCO₃ (Alfa Aesar 99.994 %) and Ta₃N₅ (made by ammonolysis of Ta₂O₅ (Aldrich 99.99 %) at 850 °C). A pellet was fired at 950 °C for 6 h and then 1500 °C for 3 h under a 5% H₂/95% N₂ gas mixture (Air Liquide). The sample was placed in a molybdenum crucible and covered with a zirconium foil to scavenge oxygen and water. Nitrogen contents from chemical analysis were found to be between 0.95(2) and 0.99(2) N atoms per formula unit without any systematic variation with Sr content x showing that the materials are essentially stoichiometric.

High resolution x-ray and neutron diffraction profiles were recorded for all of the samples at 300 K. Synchrotron X-ray powder diffraction data were measured from capillary samples at the MSPD beamline¹⁴ of the ALBA Synchrotron (Cerdanyola del Vallès, Spain). Radiation with wavelength λ = 0.6263 Å was selected by a double Si (111) crystal monochromator and calibrated with NIST Si powder. Neutron powder diffraction data were collected on the High Resolution Powder Diffractometer (HRPD) at the ISIS spallation neutron source, Rutherford Appleton Laboratory, UK. Approximately 1 g of each sample was used in a vanadium can and diffraction patterns were recorded for 2 hours. Neutron and X-ray powder data were analysed separately by Rietveld refinement using the Fullprof program.¹⁵ In the neutron analysis, structural models were refined against data collected from the backscattering bank which provides a d range of 0.65 – 2.5 Å.

Previous neutron studies have shown that SrTaO₂N has a superstructure at 300 K due to ordered rotations of the TaO₄N₂ octahedra.^{6,7} However, the x = 1 sample used in this investigation was found to have very broad superstructure peaks indicating that the rotational domains are small, and this is confirmed by electron diffraction patterns of individual microcrystallites. Further details are in ESI. Complete suppression of the superstructure was recently reported in another study where SrTaO₂N was also prepared by high temperature nitrogen treatment.¹⁶ No rotational superstructure peaks were observed in the powder neutron diffraction profiles of the other Ba_{1-x}Sr_xTaO₂N samples. Hence fits of the simple tetragonal *P4/mmm* (Fig. 1b) or cubic *Pm* $\bar{3}$ *m*

(Fig. 1c) models were used to determine the degree of anion order for all samples.

The high Δd/d resolutions of the HRPD and MSPD instruments enable the *P4/mmm* model to be tested, although the tetragonal lattice distortion is extremely small and all Ba_{1-x}Sr_xTaO₂N patterns appear cubic by eye, as shown in Fig. 2 and ESI. For x = 0.4 – 1.0 samples, the tetragonal *P4/mmm* model was found to give stable neutron refinements with consistent tetragonal strains ((a - c)/a ≈ 10⁻⁴), as shown in Fig. 3, and refined O/N site occupancies within ±7% of the ideal values for the 2D *cis*-chains model, as in Fig. 4. Hence the 2D correlated disorder previously found for x = 1 SrTaO₂N is demonstrated to extend across the x = 0.4 – 1.0 range. In contrast, neutron refinements of tetragonal lattice parameters and O/N site occupancies were unstable for the x = 0 and 0.2 samples, and so these data were fitted with the cubic *Pm* $\bar{3}$ *m* model that is consistent with 3D correlated disorder.

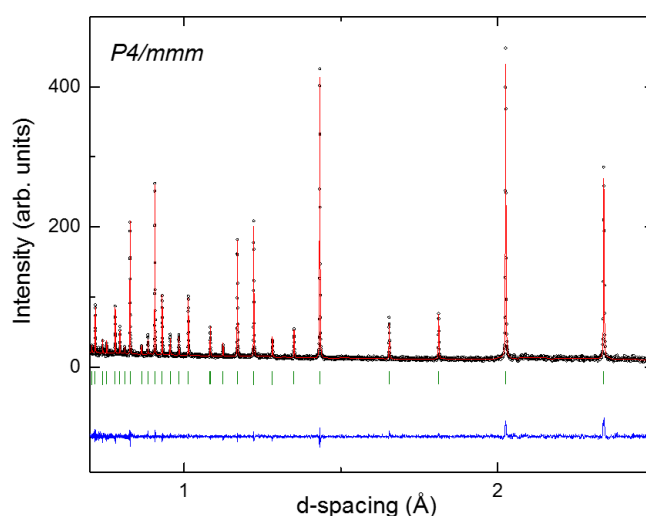


Figure 2. Fit of the tetragonal *P4/mmm* model in Fig. 1(b) describing 2D-chains to room temperature neutron powder diffraction data for Ba_{0.2}Sr_{0.8}TaO₂N. Residuals; $\chi^2 = 1.17$, $R_{wp} = 8.03\%$.

The X-ray data do not provide O/N scattering contrast but do enable the tetragonal lattice distortion to be refined. Stable refinements of a *P4/mmm* tetragonal perovskite cell were obtained for x = 0.4 – 1.0 samples, but the refinements were unstable for fits to x = 0 and x = 0.2 data, corroborating the neutron results that these two samples have true cubic *Pm* $\bar{3}$ *m* lattice symmetry. Hence both high resolution powder neutron and X-ray results demonstrate that a subtle change from cubic *Pm* $\bar{3}$ *m* to tetragonal *P4/mmm* symmetry occurs between x = 0.2 and 0.4, with similar variations in the tetragonal lattice distortion (Fig. 3), and consistent with the neutron occupancies in Fig. 4. The variations of these quantities do not correlate with the refined temperature factors shown in ESI.

The cubic Ba_{1-x}Sr_xTaO₂N structure at x = 0 – 0.2 is consistent with a 3D distribution of disordered *cis*-chains (Fig. 1e), or at least small local domains of 2D propagation distributed over the three possible orientations, while neutron occupancies show that long range 2D confinement of anion chain layers is present across the tetragonal x = 0.4 – 1 samples (Fig. 1d). The absence

of an intermediate $P4/mmm$ structure with O/N occupancies between the 2D and 3D distributions (i.e. oxygen occupancy $\approx 70-90\%$ in Fig. 4) indicates that the crossover is relatively sharp with different thermodynamic factors favouring 2D or 3D correlations.

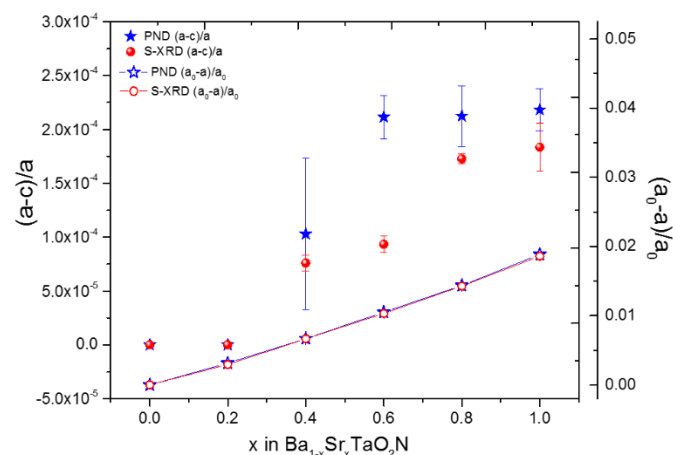


Figure 3. Variations of the tetragonal strain $(a-c)/a$ (left scale, filled points) and compressive cubic strain $(a_0 - a)/a_0$ (right scale, open points; a_0 is the lattice parameter at $x = 0$) across the $\text{Ba}_{1-x}\text{Sr}_x\text{TaO}_2\text{N}$ series with the $x = 0 - 0.2$ data from cubic perovskite structure $Pm\bar{3}m$ refinements and the $x = 0.4 - 1$ data from tetragonal perovskite structure $P4/mmm$ refinements. Red/blue points are from neutron/X-ray refinements.

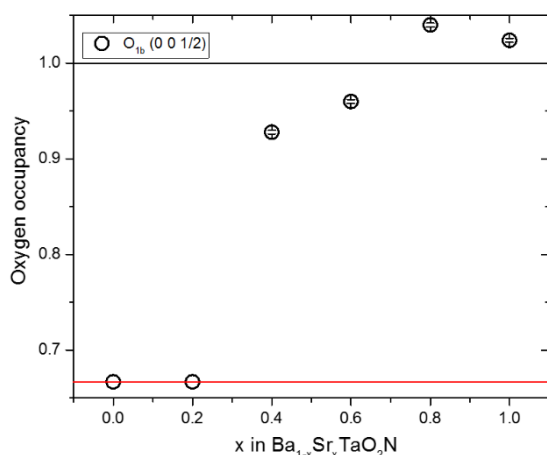


Figure 4. Plot of oxygen occupancies for the c -axis anion site from neutron refinements against x in the $\text{Ba}_{1-x}\text{Sr}_x\text{TaO}_2\text{N}$ series. Values for the $x = 0 - 0.2$ cubic $Pm\bar{3}m$ refinements are fixed at 0.67 corresponding to the model in Fig. 1c which averages over the 3D *cis*-TaN chains in Fig. 1e. Occupancies for the $x = 0.4 - 1$ samples are refined against the tetragonal $P4/mmm$ model and approximate to those shown in Fig. 1b which averages the 2D *cis*-TaN chains model in Fig. 1d.

Although both the 2D and 3D correlated arrangements (Figs. 1d and e) appear disordered by eye, previous theoretical analysis has shown that their configurational entropies are very different.¹⁷ The 2D arrangement found in $x = 1$ SrTaO_2N has a subextensive molar entropy that varies with the number of

atoms N per particle as $S = (2R \ln 2)/N^\frac{1}{2}$. The highly crystalline $\text{Ba}_{1-x}\text{Sr}_x\text{TaO}_2\text{N}$ powders used here have particle sizes $\sim 1 \mu\text{m}$ with $N \approx 1.6 \times 10^{10}$ so $S \approx 6 \times 10^{-4}R$ and hence configurational entropy is very small for the 2D materials. This unusual structural state has been described as an ‘open order’ based on the closure properties of the set of vectors for long range structural correlation.¹⁷ The 3D arrangement of chains found in $x = 0$ BaTaO_2N has a conventional configurational entropy (‘closed order’) that scales extensively, with value $S = 2R \ln(4/3) \approx 0.6R$. Hence the 2D to 3D crossover in anion chain correlation behaves thermodynamically as an order to disorder transition with an increase of $0.6R \approx 4.8 \text{ J K}^{-1} \text{ mol}^{-1}$ in configurational entropy. This must be balanced by some $\Delta H = T\Delta S$ enthalpy stabilisation of the 2D state at the crossover, and as the present samples were equilibrated at 1500°C the stabilisation of the 2D state relative to the 3D is estimated as $\Delta H \approx 8.5 \text{ kJ mol}^{-1}$. This stabilisation most likely derives from conjugation energy when all Ta-N bonds are co-planar, analogous to the stabilisation of planar structures for aromatic organic molecules, and so is very sensitive to increasing interatomic separation which lowers Ta-N orbital hybridisation. Hence the lattice expansion as x decreases (shown as a decreasing compressive cubic lattice strain on the right hand scale in Fig. 3) drives the 2D to 3D crossover in the $\text{Ba}_{1-x}\text{Sr}_x\text{TaO}_2\text{N}$ series.

In conclusion, this study demonstrates that a sharp dimensional crossover of correlated atomic disorder occurs near $x = 0.2$ in the $\text{Ba}_{1-x}\text{Sr}_x\text{TaO}_2\text{N}$ series. The 2D phase is destabilised by lattice expansion as x decreases, and the release of structural degrees of freedom provides a rapid increase in entropy due to a change from subextensive to extensive configurational entropy regimes.

We thank STFC, UK for support for H.J. and provision of ISIS beamtime, and EPSRC for additional support. This work was also supported by the Ministerio de Economía y Competitividad (MINECO), Spain through Project MAT2017-86616-R, the Severo Ochoa Program SEV- 2015-0496, and Fellowship support to AB (MAT2011-24757). We thank the ALBA synchrotron for the provision of beamtime and Prof. Rosa Palacin, Dr. Carlos Frontera (ICMAB-CSIC) and Dr. F. Fauth (ALBA) for assistance with data collection.

Conflicts of interest

There are no conflicts to declare.

Notes and references

- 1 D. A. Keen and A. L. Goodwin, *Nature* 2015, 521, 303.
- 2 Y. Tabata, H. Kadowaki, K. Matsuhira, Z. Hiroi, N. Aso, E. Ressouche and B. Fåk, *Phys. Rev. Lett.* 2006, 97, 257205.
- 3 A. Fuertes, *Mat. Horiz.* 2015, 2, 453.
- 4 A. Fuertes, *J. Mater. Chem.* 2012, 22, 3293.
- 5 J.P. Attfield, *Cryst. Growth Des.* 2013, 13, 4623.
- 6 M. Yang, J. Oró-Solé, J.A. Rodgers, A.B. Jorge, A. Fuertes and J.P. Attfield, *Nat. Chem.* 2011, 3, 47.
- 7 L. Clark, J. Oró-Solé, K. S. Knight, A. Fuertes, and J. P. Attfield, *Chem. Mater.* 2013, 25, 5004.
- 8 J. Oró-Solé, L. Clark, W. Bonin, J.P. Attfield and A. Fuertes, *Chem. Commun.* 2013, 49, 2430.
- 9 L. Clark, N. Kumar, W. Bonin, A. Sundaresan, J.P. Attfield, C. N. R. Rao and A. Fuertes. *J. Mat. Chem. C* 2014, 2, 2212.
- 10 A. B. Jorge, J. Oro-Sole, A. M. Bea, N. Mufti, T. T. M. Palstra, J.A. Rodgers, J. P. Attfield and A. Fuertes. *J. Am. Chem. Soc.*, 2008, 130, 12572.
- 11 M. Yang, J. Oro-Sole, A. Kusmartseva, A. Fuertes and J.P. Attfield. *J. Am. Chem. Soc.* 2010, 132, 4822.
- 12 H. Wolff and R. Dronskowski, *J. Comput. Chem.* 2008, 29, 2260.
- 13 K. Page, M. W. Stoltzfus, Y-I Kim, T. Proffen, P.M. Woodward, A.K. Cheetham and R. Seshadri, *Chem. Mater.* 2007, 19, 4037.
- 14 F. Fauth, I. Peral, C. Popescu, C. and M. Knapp, M. *Powder Diffraction* 2013, 28, S360.
- 15 J. Rodríguez-Carvajal, Commission on Powder Diffraction (IUCr) Newsletter 2001, 26, 12.
- 16 D. Chen, D. Habu, Y. Masubuchi, S. Torii, T. Kamiyama and S. Kikkawa, *Solid State Sci.* 2016, 54, 2.
- 17 P.J. Camp, A. Fuertes and J.P. Attfield. *J. Am. Chem. Soc.*, 2012, 134, 6762.